

Application of SQUIDs to phase sensitive experiments (in the $\text{Sr}_{1-x}\text{La}_x\text{CuO}_2$ system)

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Outline

Motivation behind this work

SQUIDs (Superconducting Quantum Interference Devices) – brief description

The $\text{Sr}_{1-x}\text{La}_x\text{CuO}_2$ (SLCO) superconducting compounds and microstructural properties of the SLCO thin films grown by PLD on $\text{BaTiO}_3/\text{SrTiO}_3$

$\text{Sr}_{1-x}\text{La}_x\text{CuO}_2$ ($x=0.145$) 30° [001] tilt grain boundary junctions

Symmetry of the superconducting order parameter symmetry for $\text{Sr}_{1-x}\text{La}_x\text{CuO}_2$ ($x=0.145$) compounds from phase sensitive experiments using 0- and π -dc SQUIDs

Conclusions

Motivation of the work

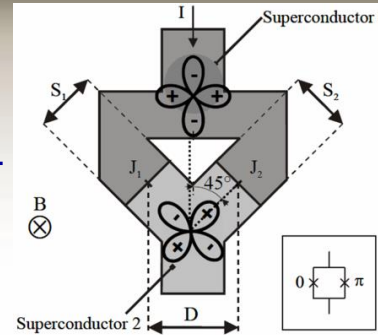
1. Symmetry of the superconducting order parameter in the electron-doped $\text{Sr}_{1-x}\text{La}_x\text{CuO}_2$ compounds (SLCO, $x=0.10-0.175$) from:

i. tunneling studies across Josephson junctions (grain boundary, ramp-type, planar):

Tomaschko et al., PRB 84, 0214507 (2011)

ii. phase sensitive tunneling experiments based on $0/\pi$ -SQUIDS (on SrTiO_3 tetracrystals)

Tomaschko et al., PRB 86, 094509 (2012)



2. Studies on SLCO Josephson junctions:

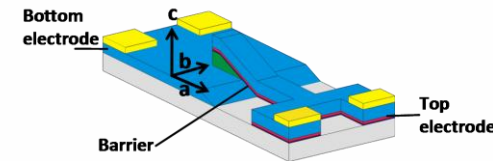
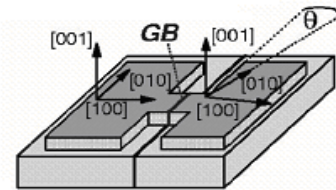
i. grain-boundary junctions based on c-axis oriented SLCO films

Tomaschko et al., PRB 84, 0214507 (2011)

ii. ramp type SLCO/Me/Nb (Me=Ag, Au) junctions

iii. planar c-axis SLCO/Au/Nb tunnel junctions

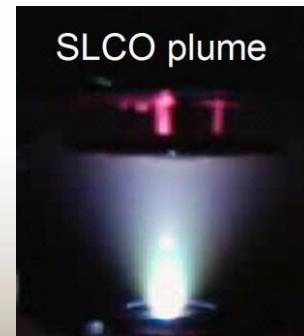
Tomaschko et al., PRB 84, 064521 (2011)



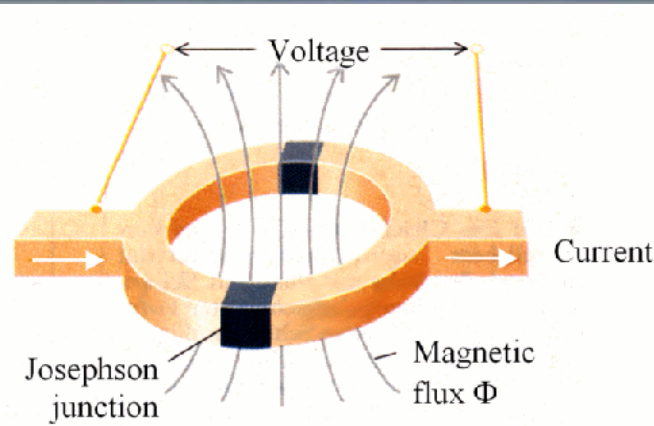
3. Correlation between strain, microstructure, and electrical transport properties in SLCO thin films grown by PLD on different substrates (SrTiO_3 , KTaO_3 , and DyScO_3)

Leca et al., APL 89, 92504 (2006); Appl. Phys. A 93, 779 (2008)

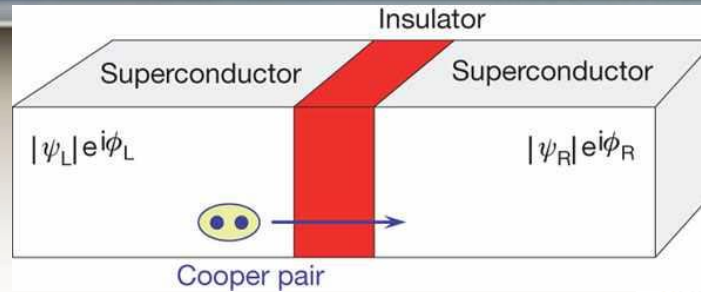
Tomaschko et al., PRB 85, 024519 (2012)



SQUIDs and Josephson junctions— brief description

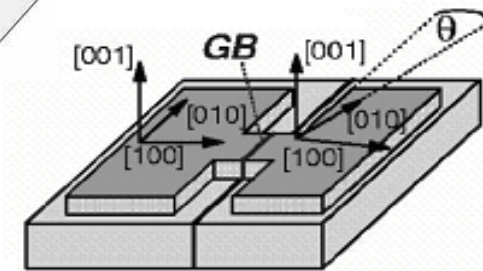


Design of a dc-SQUID

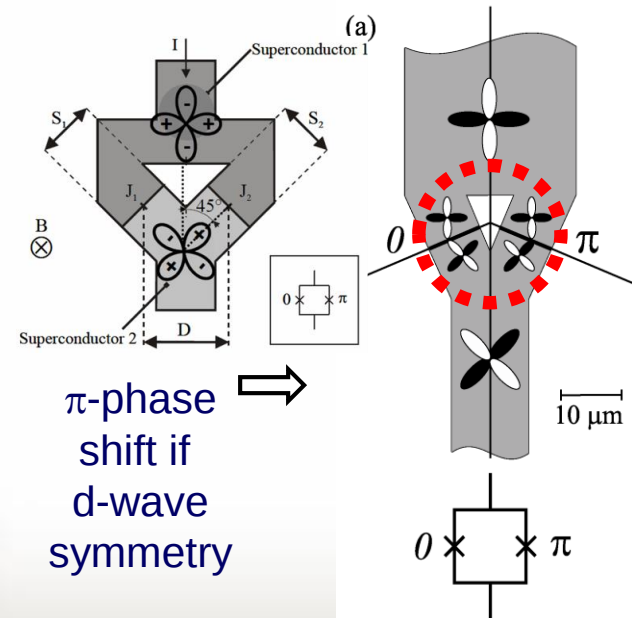


Josephson junction

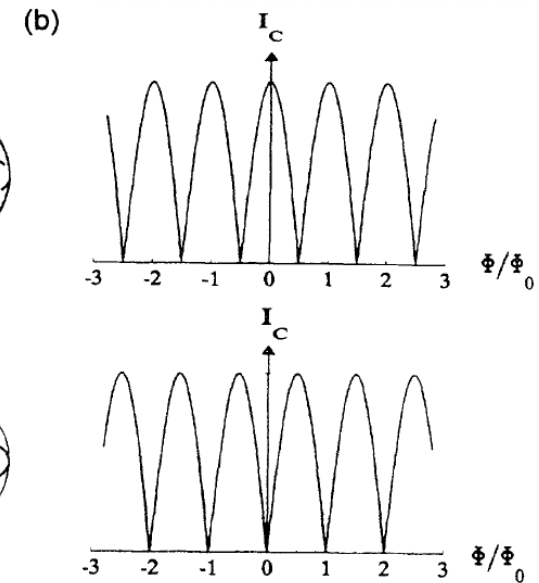
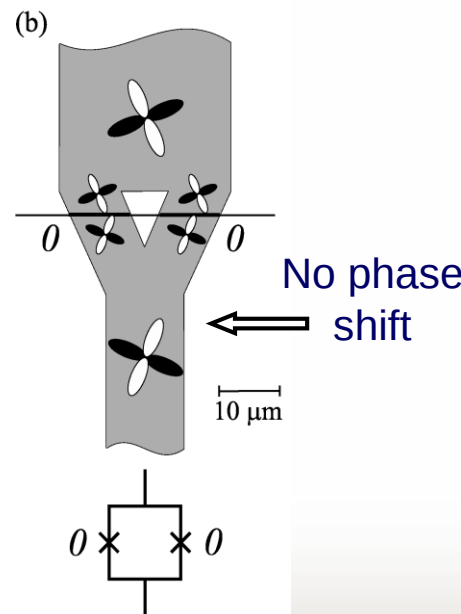
Grain-boundary Josephson junction



Phase sensitive tunneling:

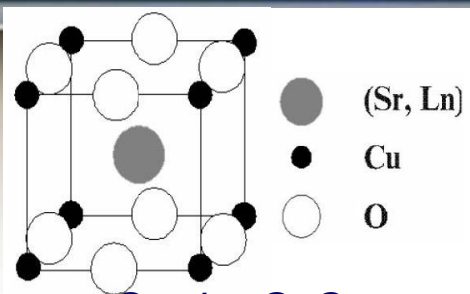


0- and π -SQUIDS



I_c vs applied field for s- and d-wave superconductors

The $\text{Sr}_{1-x}\text{La}_x\text{CuO}_2$ compounds



-infinite-layer (IL) type structure

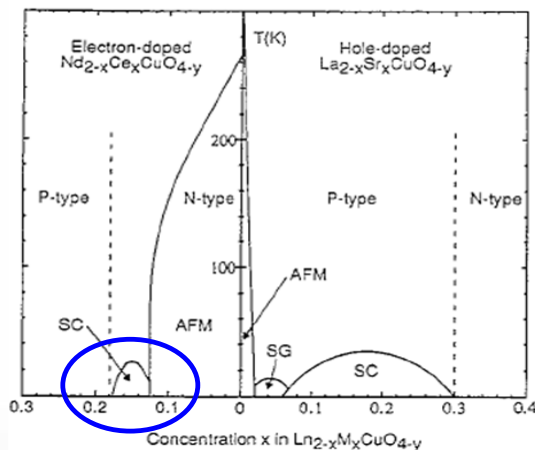
- $T_{c,\text{max}} = 43 \text{ K}$

Smith et al., Nature 351 (1991)

Adachi et al., Physica C 196 (1992)

ELECTRON-HOLE SYMMETRY (QUALITATIVE)

Metallic ← Insulating → Metallic

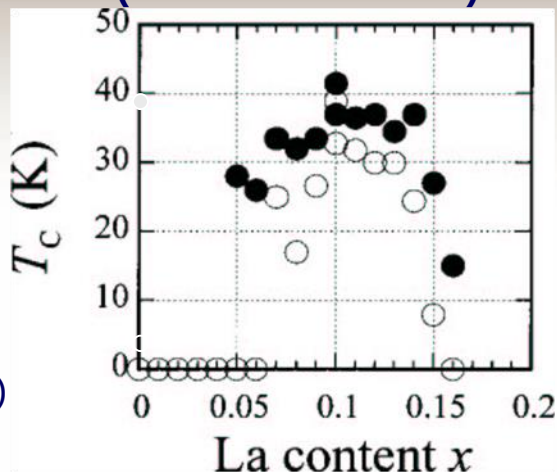


Phase diagram for $T'\text{-Ln}_{2-x}\text{Me}_x\text{CuO}_4$

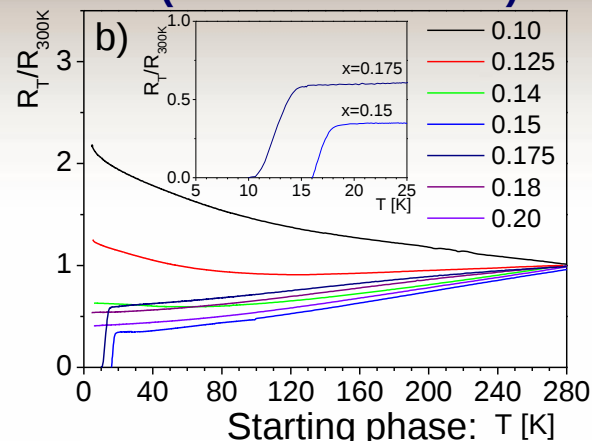
Armitage et al., RMP 82 (2010)

Unknown phase diagram for IL phases

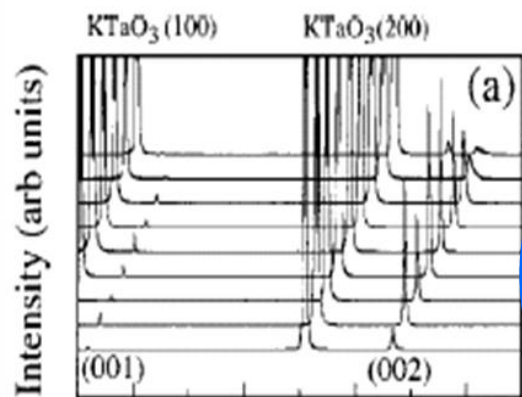
**MBE growth on (001) KTaO_3
(reduction method)**



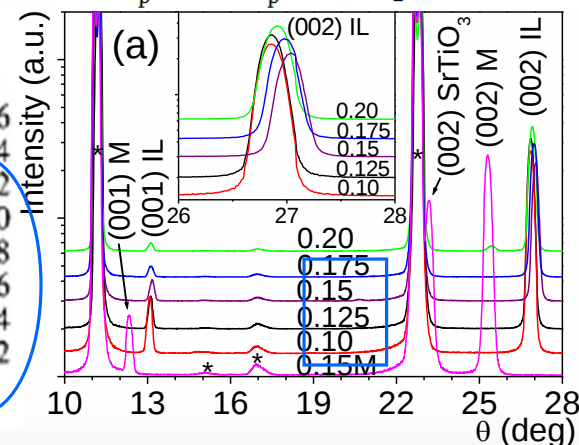
**PLD growth on (001) KTaO_3
(oxidation method)**



$2\sqrt{2}a_p \times 2\sqrt{2}a_p \times c$ (super)structure



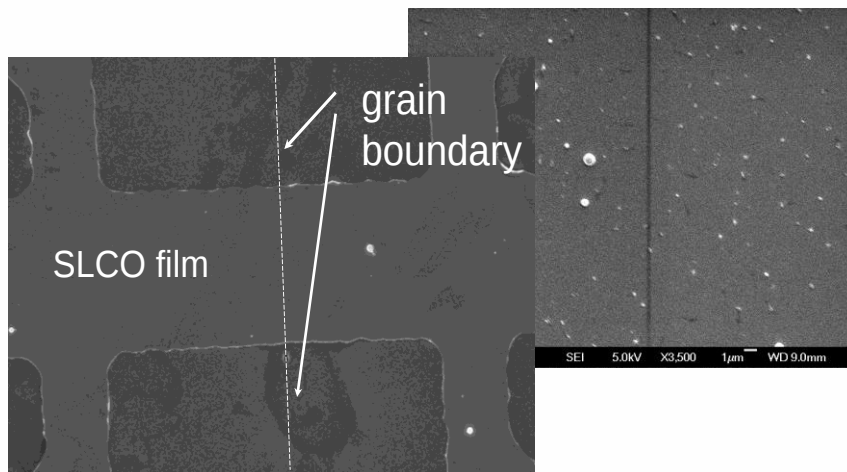
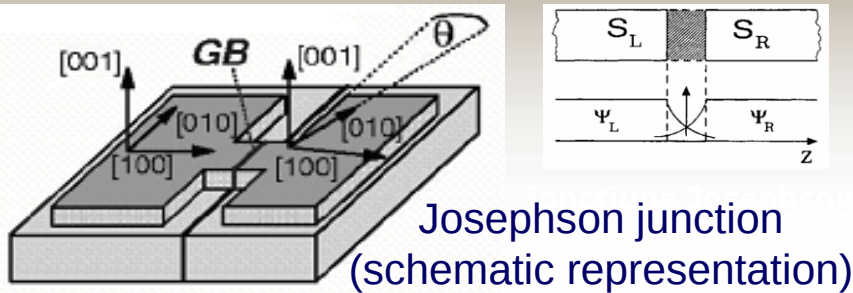
Karimoto et al.
Appl. Phys. Lett. 79 (2001)



Leca et al.
Appl. Phys. Lett. 89 (2006)

Requirements for GB junctions and films growth parameters

Grain-boundary junctions



SEM of SLCO grain-boundary junction

Single phase, c-axis oriented films required

On BaTiO_3 -buffered (001) SrTiO_3 BaTiO_3 (BTO):

$T_d = 700\text{-}850\text{ }^\circ\text{C}$

$P_d = 0.10\text{ mbar O}_2$

Post-deposition annealing:

-15 min @ 950°C , under 0.10 mbar O_2 , and
30 min @ 950°C , in vacuum (10^{-7} mbar)

$\text{Sr}_{1-x}\text{La}_x\text{CuO}_2$ (SLCO):

$T_d = 550\text{-}600\text{ }^\circ\text{C}$

$P_d = 0.20\text{ mbar O}_2$

Post-deposition annealing (reduction):

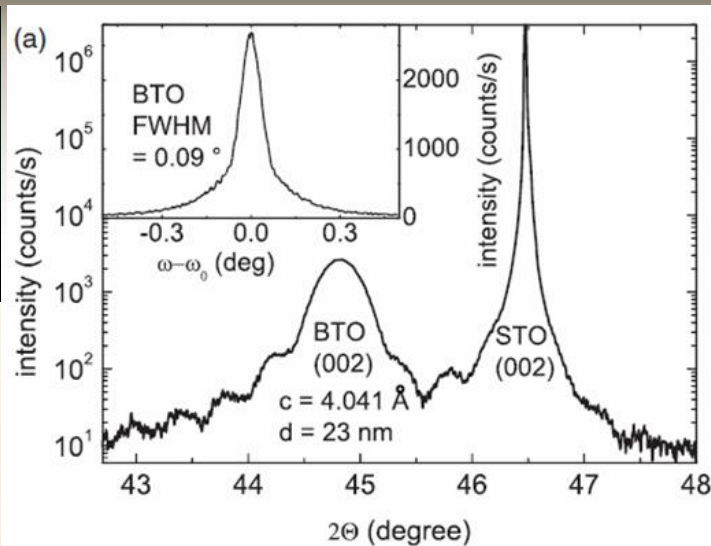
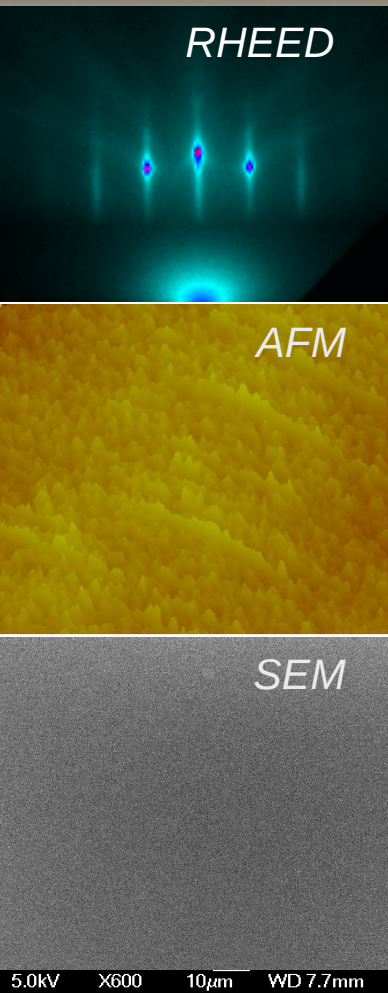
45-60 min @ $550\text{ }^\circ\text{C}$, in vacuum (10^{-7} mbar)

T_c : up to 21 K

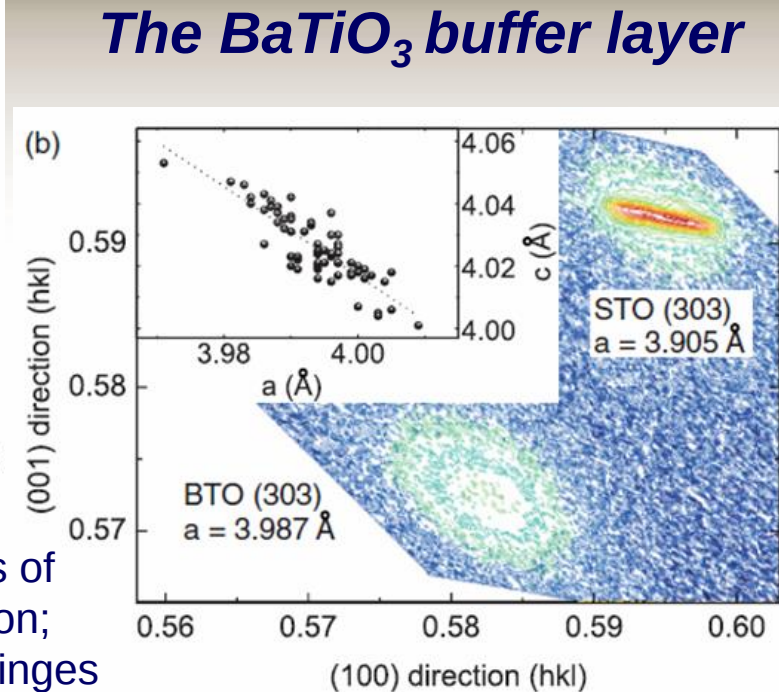
Film's composition ($x_t = 0.15$ target),
from RBS data: $x_f = 0.145$

$a_{\text{STO}} = 3.905\text{ \AA}$, $a_{\text{BTO}} \sim 3.987\text{ \AA}$, $a_{\text{SLCO}} \sim 3.967\text{ \AA}$

$\text{Sr}_{1-x}\text{La}_x\text{CuO}_2$ ($x \approx 0.15$) grown on $\text{BaTiO}_3/\text{SrTiO}_3$ (001)



XRD $2\theta/\omega$ and rocking curve scans of BTO/STO symmetric (002) reflection; film thickness estimated from Laue fringes



The BaTiO_3 buffer layer

XRD rsm around asymmetric (303) Bragg reflection of BTO/STO

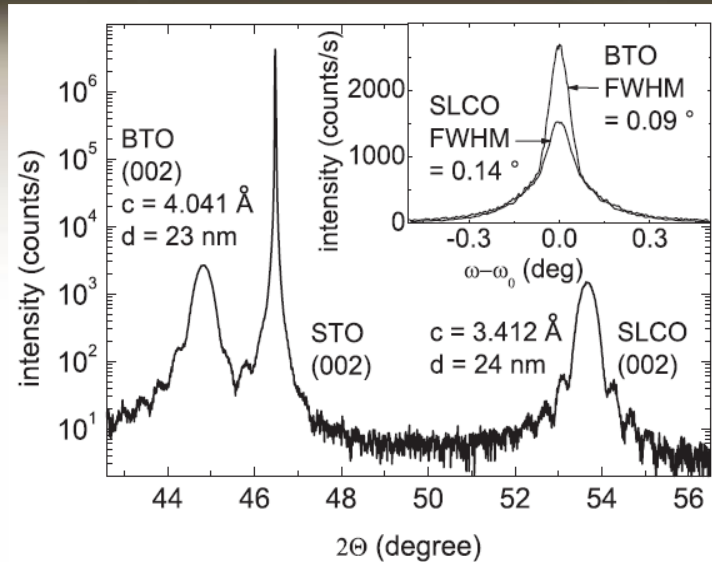
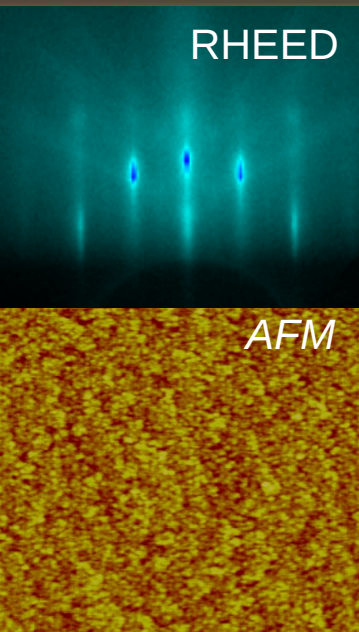
Tetragonal symmetry

Deposition conditions for BaTiO_3 :

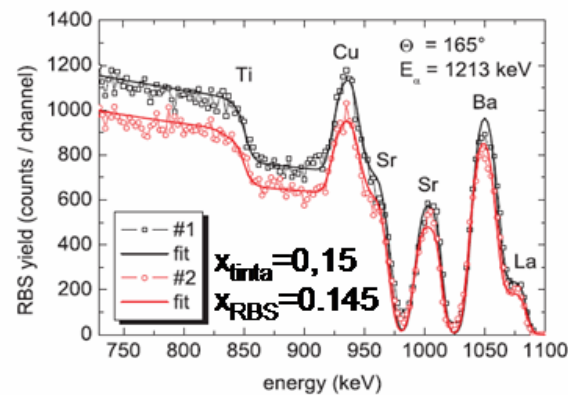
$T_d = 750^\circ\text{C}$, $P_d = 0.10 \text{ mbar O}_2$, $E_d = 1.75 \text{ J/cm}^2$;

annealing 15 min/ 950°C / 0.10 mbar O_2 and 30 min/ 950°C / 10^{-7} mbar

$\text{Sr}_{1-x}\text{La}_x\text{CuO}_2$ ($x \approx 0.15$) grown on $\text{BaTiO}_3/\text{SrTiO}_3$ (001)

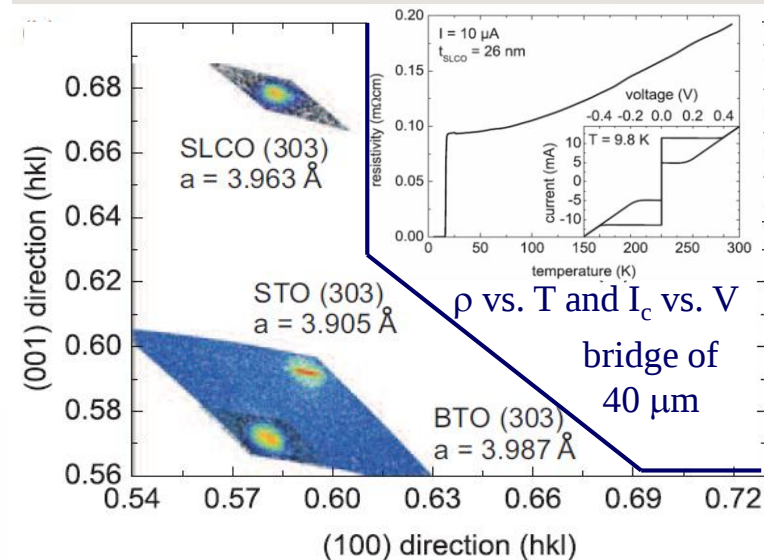


XRD $2\theta/\omega$ and rocking curve scans of (002) reflections



RBS data

The $\text{Sr}_{1-x}\text{La}_x\text{CuO}_2$ films



XRD rsm of (303) SLCO/BTO/STO

No presence of secondary phase; tetragonal, IL-type structure; preferential orientation (c-axis)

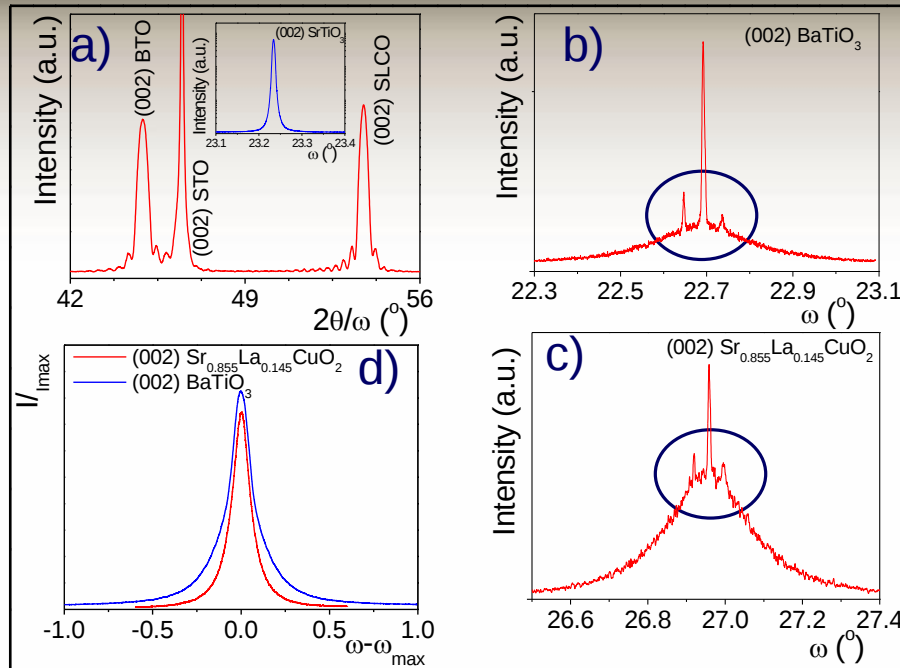
Deposition conditions for $\text{Sr}_{1-x}\text{La}_x\text{CuO}_2$:

$T_d = 550^\circ\text{C}$, 0.30 mbar O_2 ; annealing 50 min/ $550^\circ\text{C}/10^{-7}$ mbar

$T_{c,0} \leq 21 \text{ K}$; $J_{c@4.2\text{K}} = 2.1 \times 10^6 \text{ A/cm}^2$

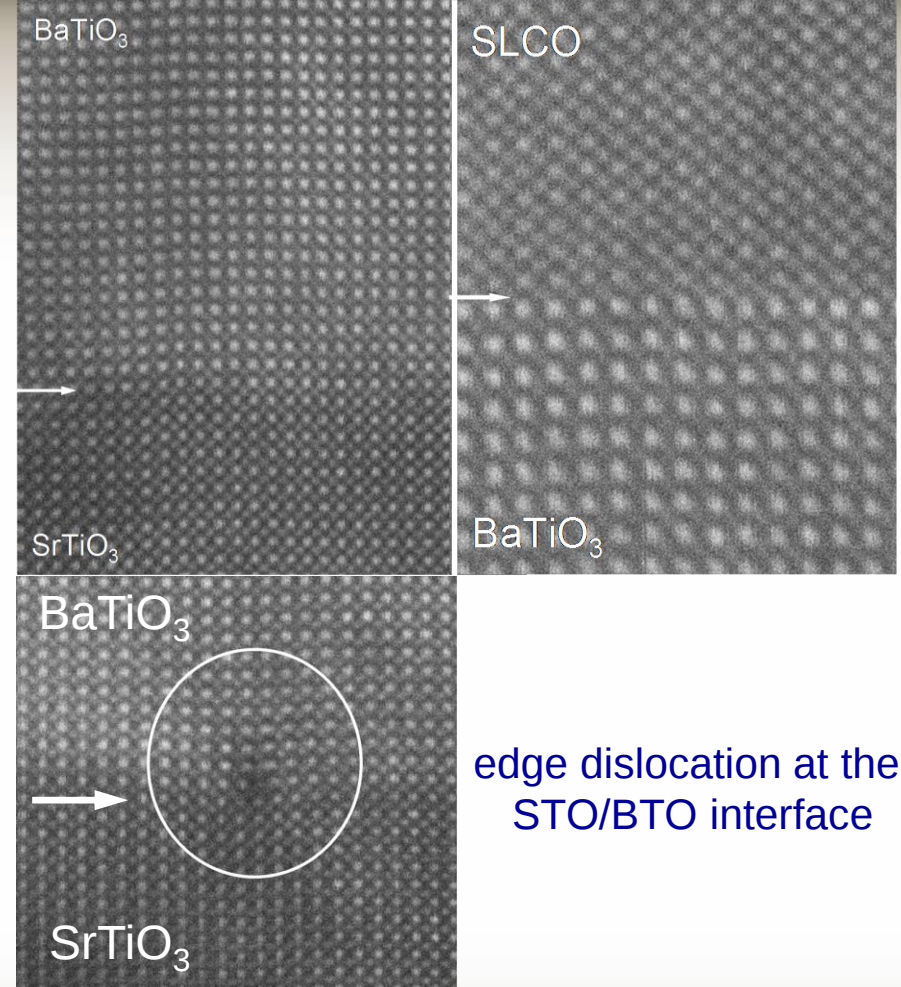
$\text{Sr}_{1-x}\text{La}_x\text{CuO}_2$ ($x \approx 0.15$) grown on $\text{BaTiO}_3/\text{SrTiO}_3$ (001)

HRXRD



Microstructure

HRSTEM



↑ HRXRD (a-c) and XRD (d) data for a BTO/SLCO film with $d_{\text{BTO}} = 35$ nm
Misfit strain accommodation by lattice modulations
Two components in ω scans: strained and relaxed

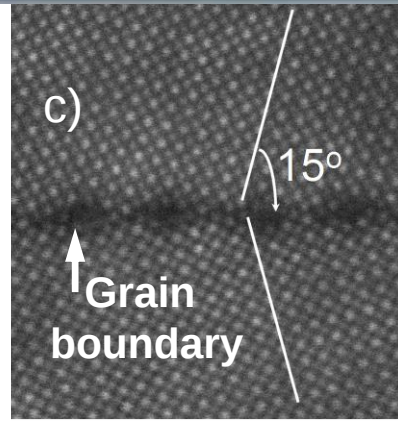
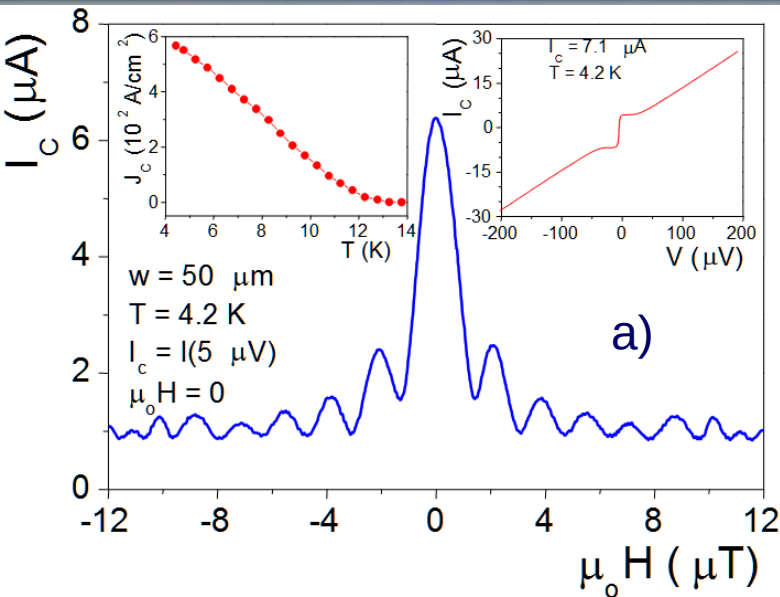
Buffer layer thickness dependent microstructure:
 - lattice modulations visible **only** in the HRXRD scans and for $d_{\text{BTO}} > 30$ nm

HRXRD: triple axes configuration
 XRD: double axes configuration

Migration of the structural defects at the BTO/STO interface due to high temperature annealing

Tera et al., APL 80 (2002)

$\text{Sr}_{1-x}\text{La}_x\text{CuO}_2$ 30° symmetric grain boundary junctions



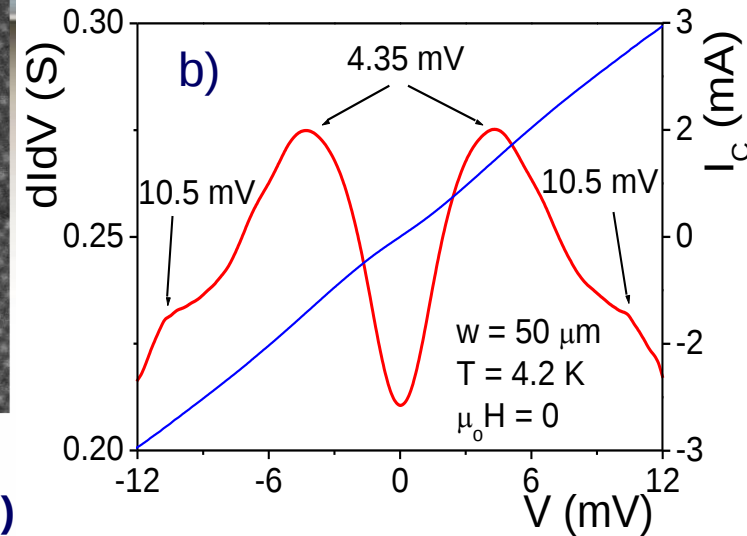
HRSTEM image
(SrTiO_3 bicrystal)

$I_c H$, $J_c T$, and $I_c V$ characteristics

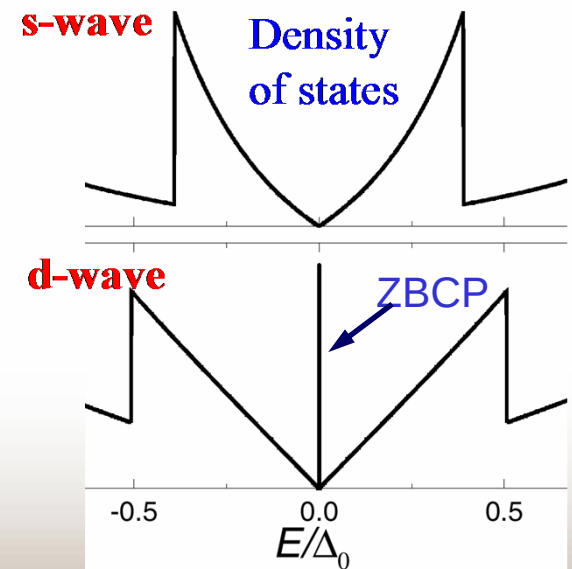
- a) $I_c V$ s: resistively and capacitively shunted junction (RCSJ)-like;
- highly regular Fraunhofer-like patterns;
 - high J_c @ 4.2 K of $\sim 0.55 \text{ kA/cm}^2$; $I_c R_n$ of $\sim 50 \mu\text{V}$

- b) conductance spectra did not show a zero-bias conductance peak-ZBCP (but a V-shaped subgap spectra) **and** extra peaks above the coherence peaks; microscopic roughness?
- possible first observation of the two band structure (due to holes and electrons charge carriers);

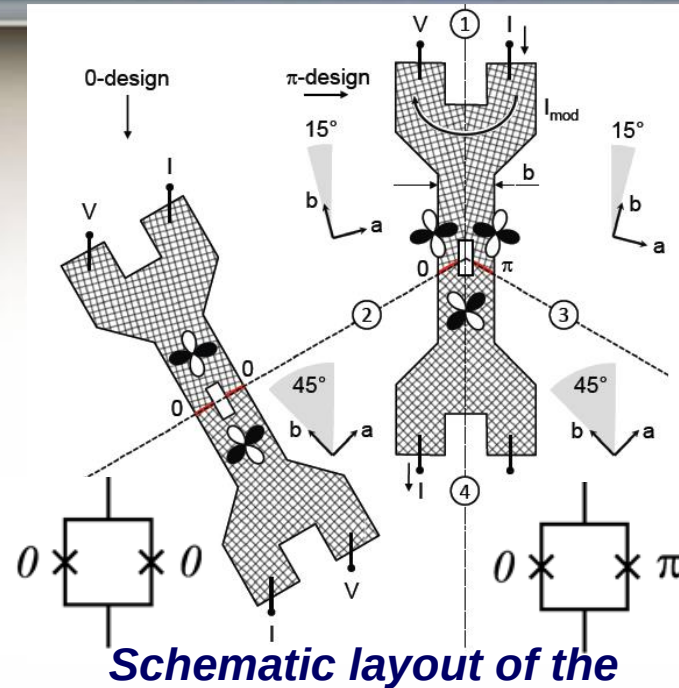
- c) plain view HRTEM of the substrate (30°) grain boundary.



Conductance spectra



Pairing symmetry from phase sensitive experiments



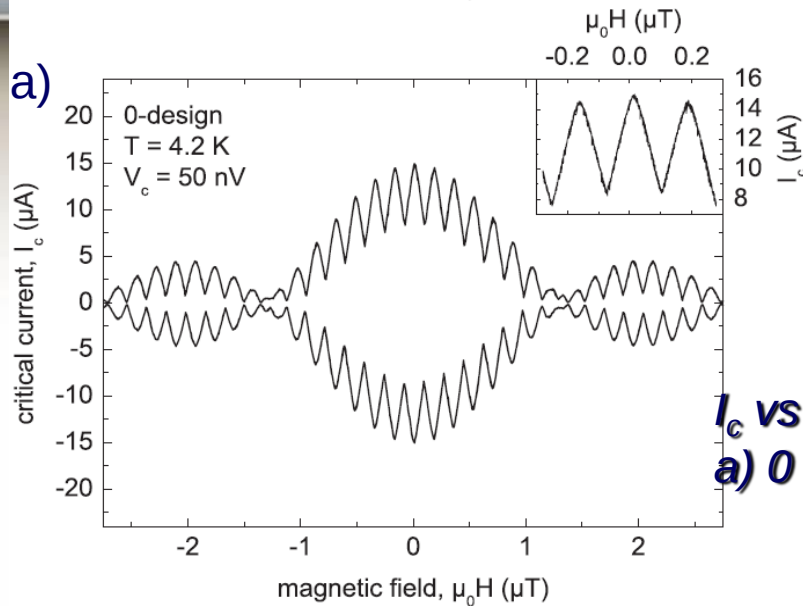
[Schulz et al., APL 76 (2000); Chesca et al., PRL 90 (2003)]

Geometry designed to be frustrated for d-wave pairing;
If d-wave: the dc-SQUID ring around the tetracystal point contains one 0-junction and one π -junction (forming the π -SQUID, which exhibits an additional π shift in its phase).

The I_c vs. H shows a minimum at $H=0$ for the π -SQUID due to the phase shift across the π -junction.

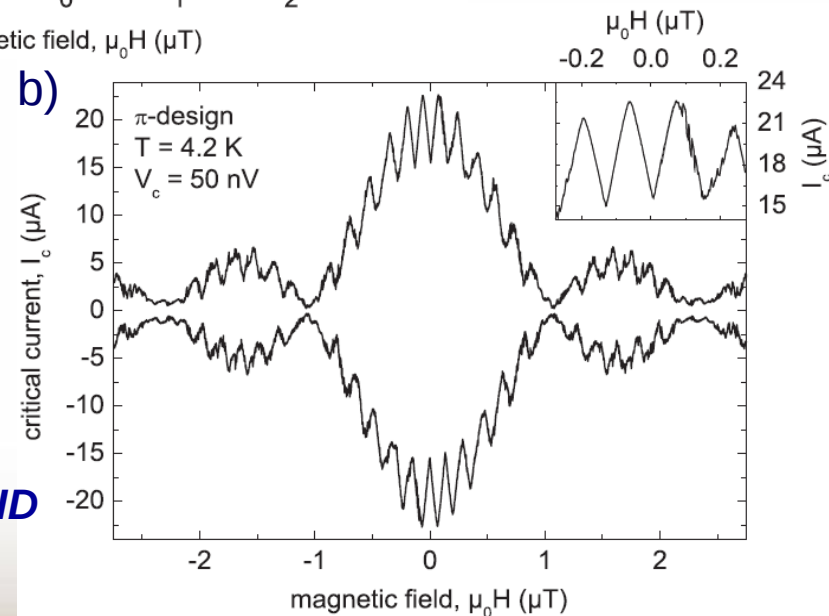
Predominantly $d_{x^2-y^2}$ pairing symmetry

a)



I_c vs magnetic field for a) 0 - and b) π -SQUID

b)

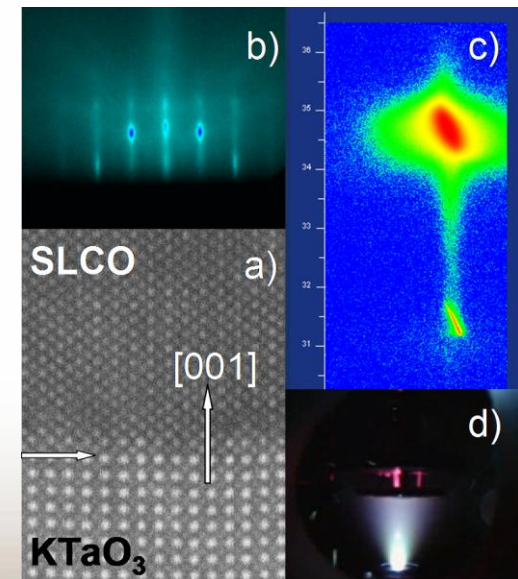
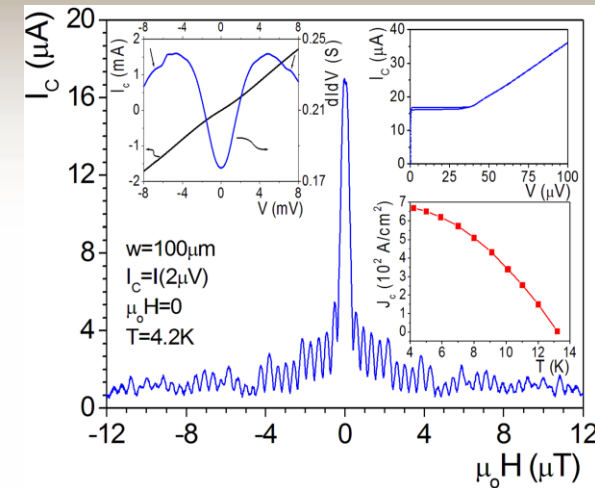


Tomaschko et al., Phys. Rev. B 86 (2012)

Electroceramics XIV 17.06.2014–11/12

Conclusions

- no zero bias conductance peak observed in grain-boundary junctions, most probably due to oxygen vacancies along the grain boundary;
- phase sensitive experiments shown a predominantly d-wave order parameter symmetry in the $\text{Sr}_{0.855}\text{La}_{0.145}\text{CuO}_2$ thin films;
- the superconducting properties of the PLD grown electron-doped $\text{Sr}_{1-x}\text{La}_x\text{CuO}_2$ thin films are still far of those of the bulk or of the MBE grown films; however, higher phase stability can be achieved by PLD (in the over doped region).



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