

# *Structural and electrical properties in superconducting $Sr_{0.85}La_{0.15}CuO_2$ -based nanostructures*

**Victor Leca<sup>1,2,3</sup>, Jochen Tomaschko<sup>2</sup>,  
Wim Arnold Bik<sup>4</sup>, Reinhold Kleiner<sup>2</sup>, Dieter Kölle<sup>2</sup>**

<sup>1</sup>*National Institute for Research and Development in Microtechnologies, Bucharest, Romania*

<sup>2</sup>*Center for Collective Quantum Phenomena in LISA+, Universität Tübingen, Germany*

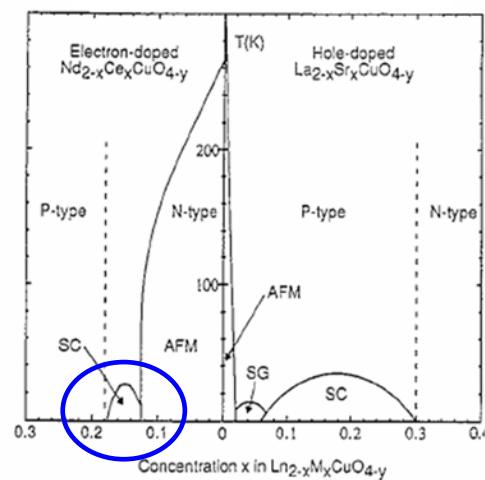
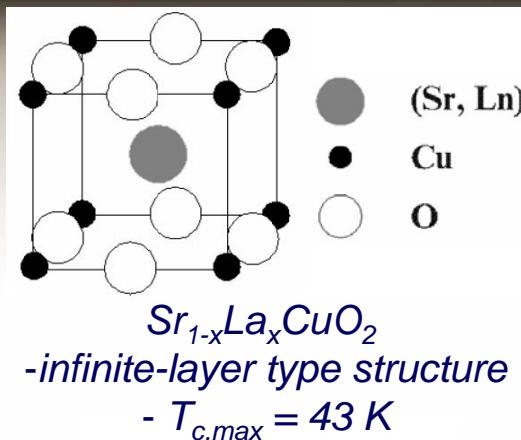
<sup>3</sup>*Faculty of Applied Chemistry and Materials Science, Polytechnique University of Bucharest, Romania*

<sup>4</sup>*AccTec BV, TN/Cyclotrongebouw, Technische Universiteit Eindhoven, The Netherlands*

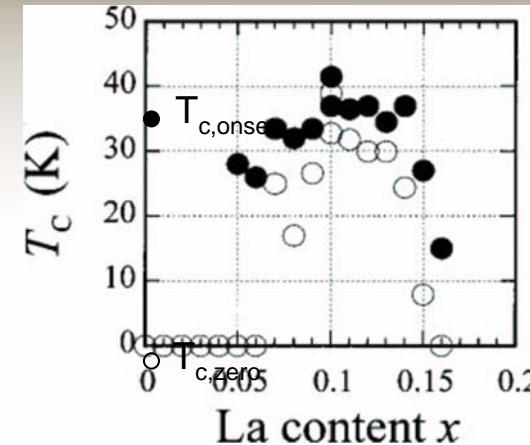
# Outline

1. Experimental motivation of the work - the electron-doped infinite-layer type  $\text{Sr}_{1-x}\text{La}_x\text{CuO}_2$  high  $T_c$  superconductors
2.  $\text{Sr}_{1-x}\text{La}_x\text{CuO}_2$  ( $x=0.10-0.20$ ) thin films: PLD growth by the reduction method and characterization
3. Grain boundary  $\text{Sr}_{1-x}\text{La}_x\text{CuO}_2$  ( $x=0.15$ ) Josephson junctions
4. Preliminary results on the  $\text{Sr}_{1-x}\text{La}_x\text{CuO}_2/\text{Au}/\text{Nb}$  ramp-type junctions
5. Conclusions

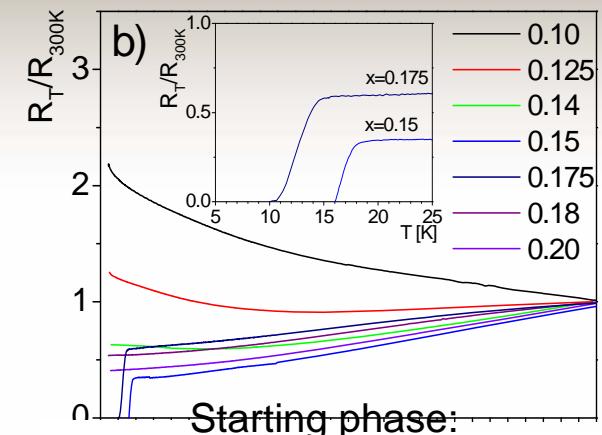
# The $\text{Sr}_{1-x}\text{La}_x\text{CuO}_2$ (SLCO) compounds



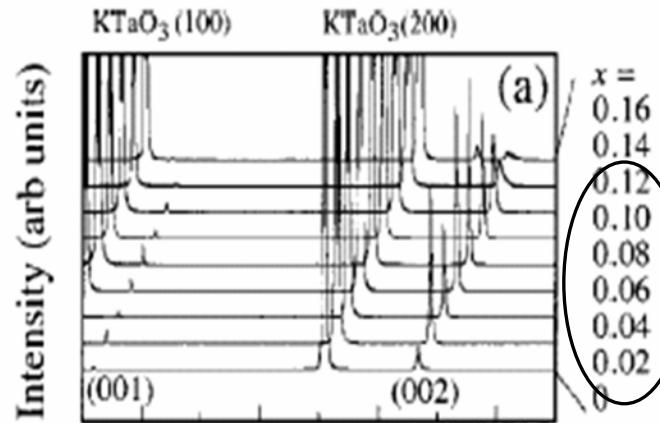
*MBE growth on (001)  $\text{KTaO}_3$   
(reduction method)*



*PLD growth on (001)  $\text{KTaO}_3$   
(oxidation method)*



$2\sqrt{2}a_p \times 2\sqrt{2}a_p \times c$  (super)structure



Karimoto et al.

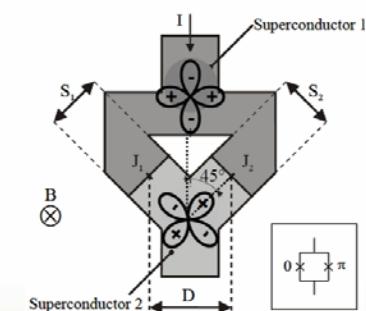
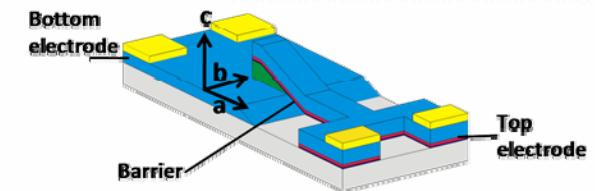
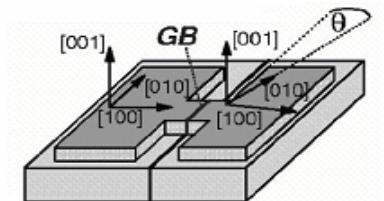
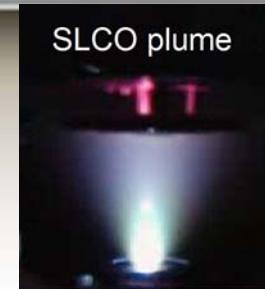
*Appl. Phys. Lett.* 79 (2001)

Leca et al.

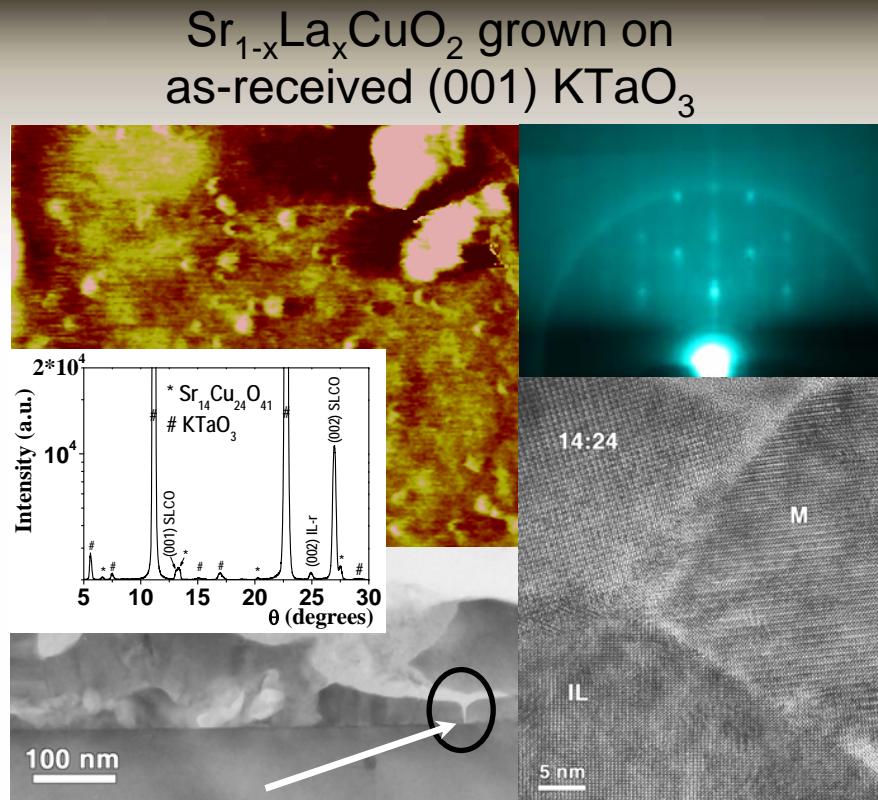
*Appl. Phys. Lett.* 89 (2006)

# Motivation of the work

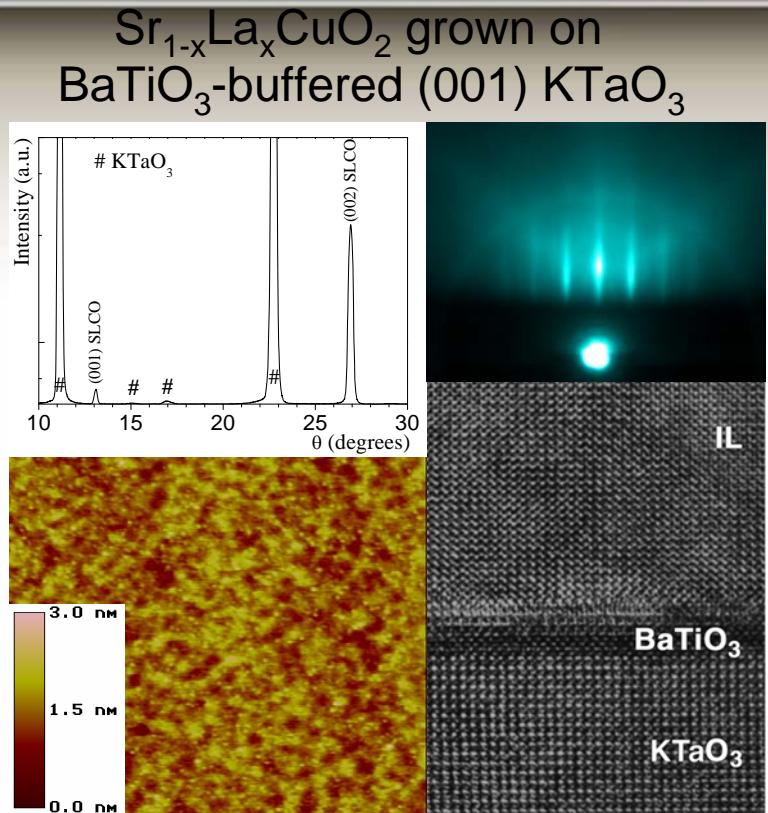
1. Studies on the  $\text{Sr}_{1-x}\text{La}_x\text{CuO}_2$  ( $x=0.10-0.175$ ) thin films: growth and characterization; PLD growth of SLCO thin films on different substrates: (001)  $\text{SrTiO}_3$ , (001)  $\text{KTaO}_3$ , and (110)  $\text{DyScO}_3$  – role of the epitaxial strain on structural and transport properties
2. Fabrication of the grain-boundary Josephson junctions on  $\text{BaTiO}_3$  buffered  $\text{SrTiO}_3$  and studies of their transport properties (for  $x=0.125-0.15$ )
3. Development of the ramp-type  $\text{Sr}_{1-x}\text{La}_x\text{CuO}_2/\text{Me}/\text{Nb}$  ( $\text{Me}=\text{Ag}, \text{Au}$ ) Josephson junction technology and studies on junctions transport properties (for  $x>0.15$ )
4. Studies on the order parameter symmetry – based on GB (tetracystal) or ramp-type Josephson junctions for different La doping levels ( $x=0.10-0.175$ ) and temperatures ( $2 \text{ K}-T_c$ )



# Role of the substrate-film interface

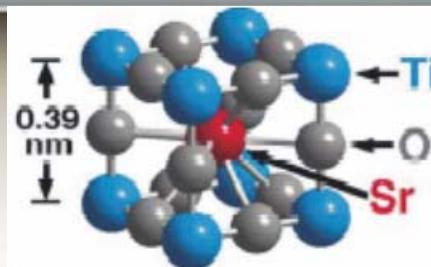
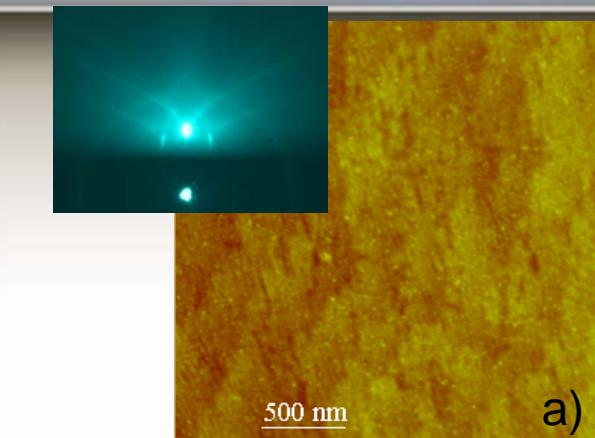


XRD, HRTEM: Structural defects/secondary phases developing at the substrate-film interface  
RHEED, AFM: 3D growth mode induced by the high interface roughness

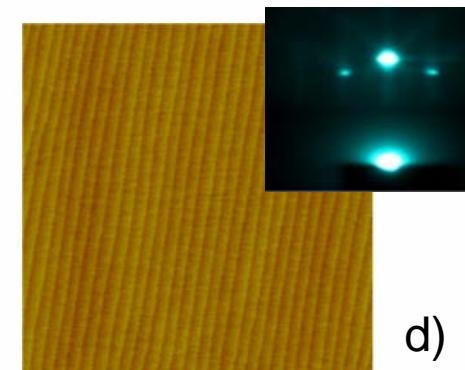
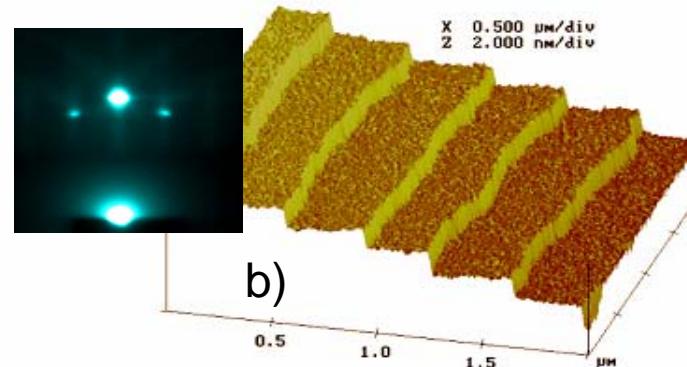
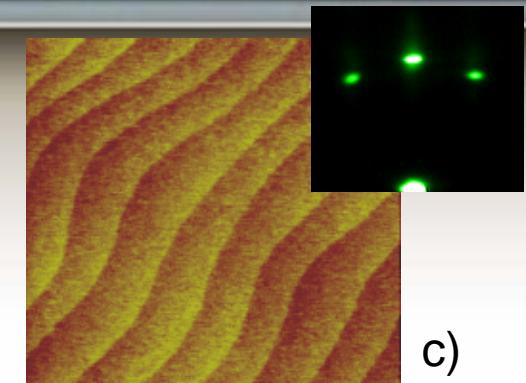


XRD, HRTEM, AFM: single-phase, lower density of structural defects, improved morphology  
RHEED, AFM: 2D growth mode

# Substrate morphology – $\text{SrTiO}_3$



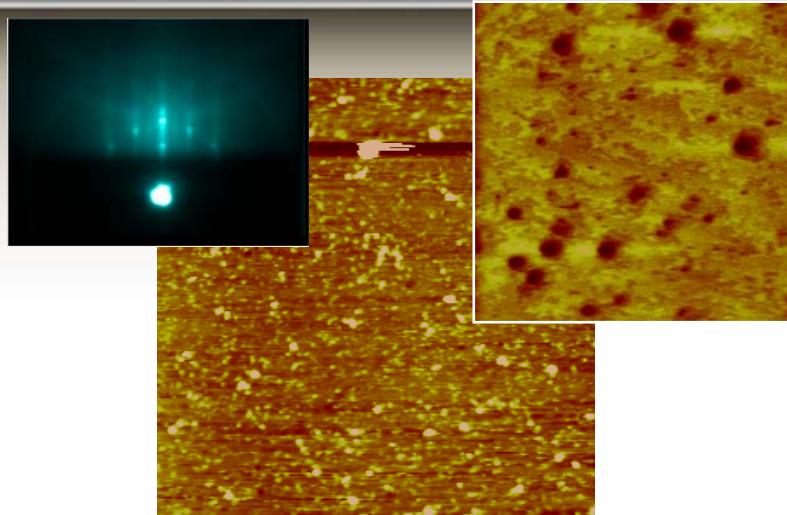
$\text{SrTiO}_3$  structure



- a) as-received substrate
- b) after chemical treatment (BHF) and annealing ( $950^\circ\text{C}/1\text{h}/\text{O}_2$ ) -  $\text{TiO}_2$  termination
- c) after annealing ( $950^\circ\text{C}/1\text{h}/\text{O}_2$ ) -  $\text{SrO}$  termination
- d) after chemical treatment (BHCl) and annealing ( $950^\circ\text{C}/1\text{h}/\text{O}_2$ ) -  $\text{TiO}_2$  termination  
(high miscut angle case)

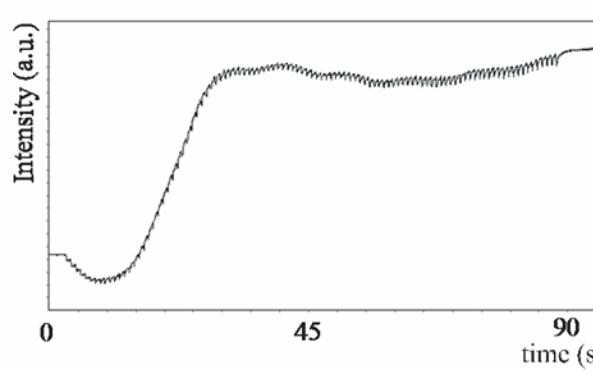
Kawasaki et al. *Science* 226 (1994)  
Koster et al. *Appl. Phys. Lett* 73 (1998)  
Leca, *PhD thesis* (2003)

# Substrate morphology – $\text{KTaO}_3$



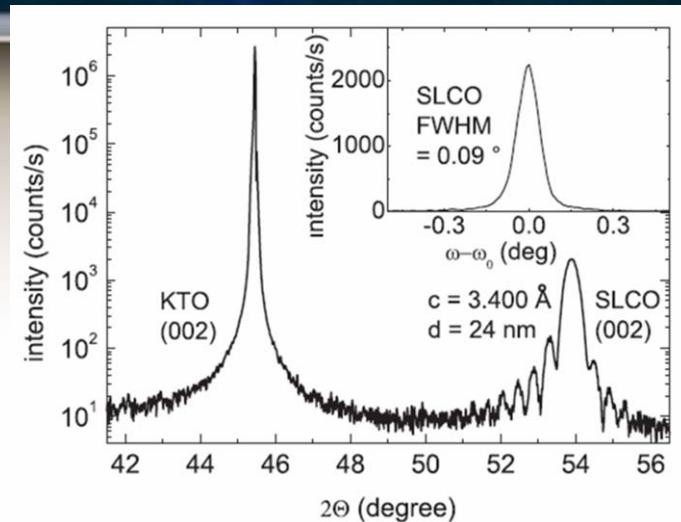
*as-received substrate*

**$\text{KTaO}_3$ : new methods required**



*“surface healing”:  $\text{BaTiO}_3$  on  $\text{KTaO}_3$  –  
RHEED intensity evolution and AFM of the resulted surface*

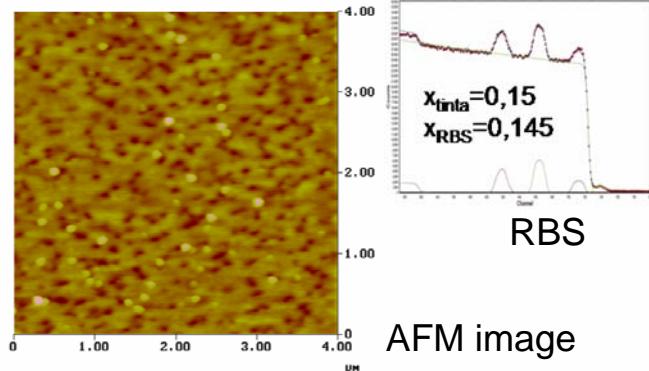
# $Sr_{1-x}La_xCuO_2$ ( $x=0.15$ ) grown on $KTaO_3$ (001)



2 $\theta$ /ω XRD scan of (002) SLCO/KTO

$$a_{SLCO} \sim 3.978-3.988 \text{ \AA}$$

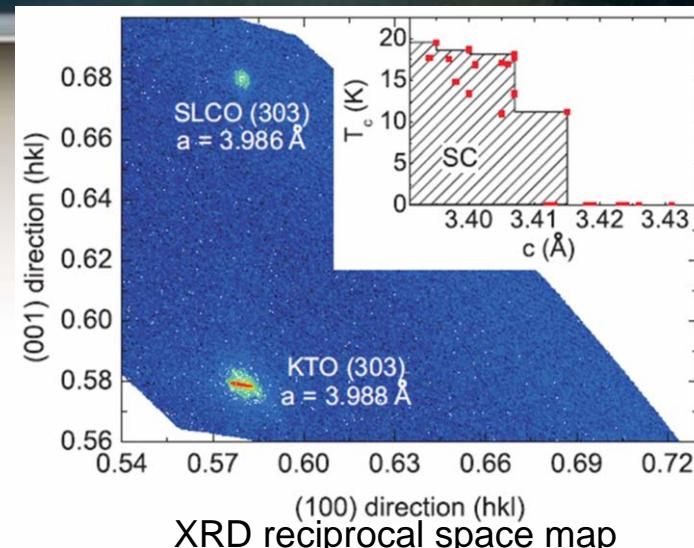
$$c_{SLCO} \sim 3.395-3.405 \text{ \AA}$$



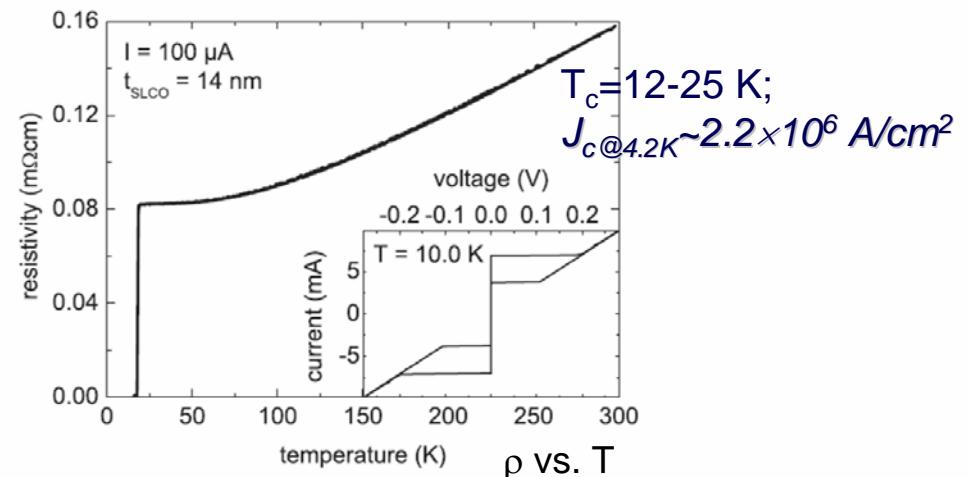
AFM image

Growth parameters:

$Sr_{0.85}La_{0.15}CuO_2$  (SLCO):  $T_d = 575^\circ\text{C}$ ,  $P_d = 0.30 \text{ mbar O}_2$ ;  $E_d = 2 \text{ J/cm}^2$ ; 20 min/550°C/10<sup>-7</sup> mbar  
films thickness: 15-50 nm;



XRD reciprocal space map

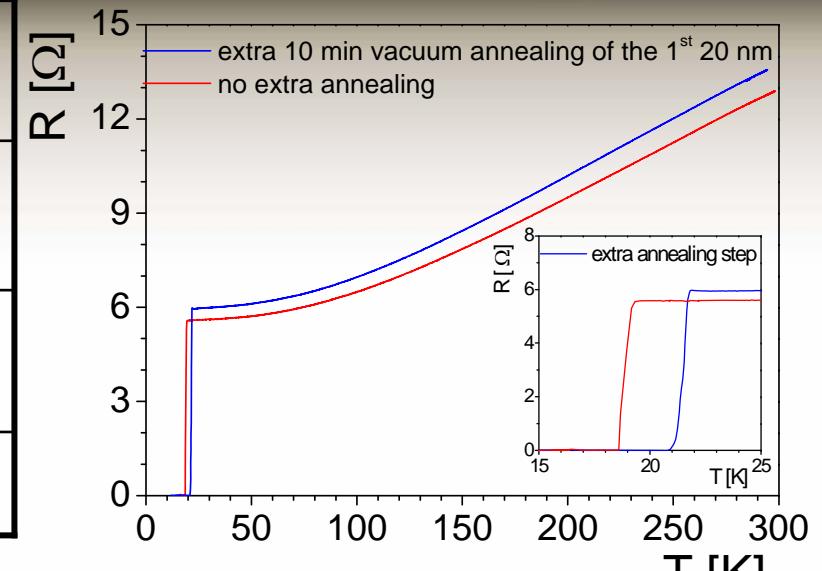
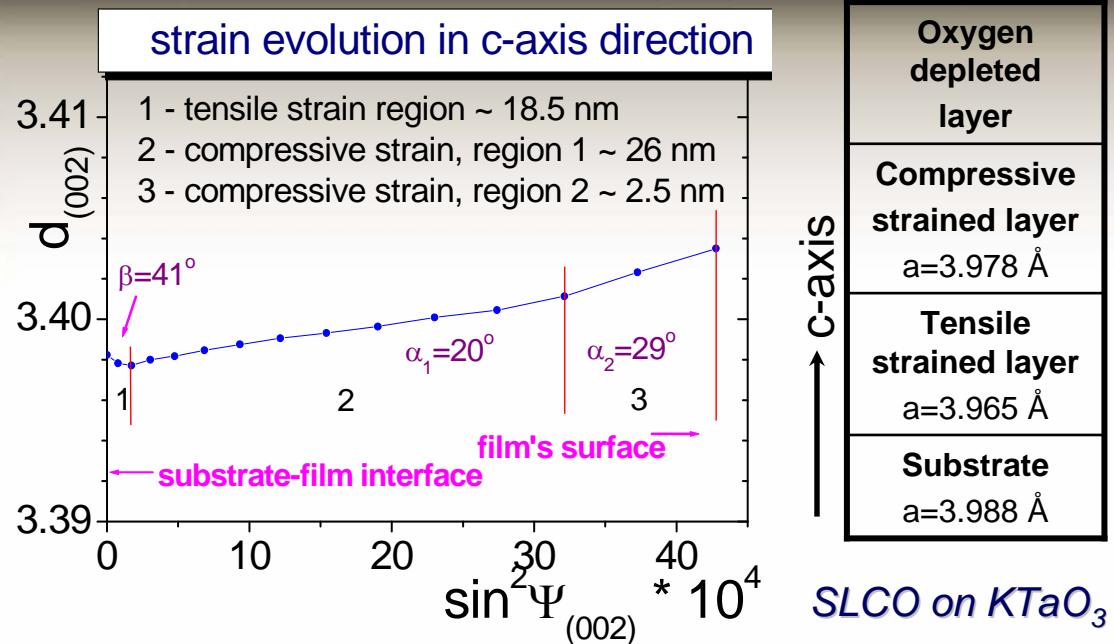


$$T_c = 12-25 \text{ K};$$

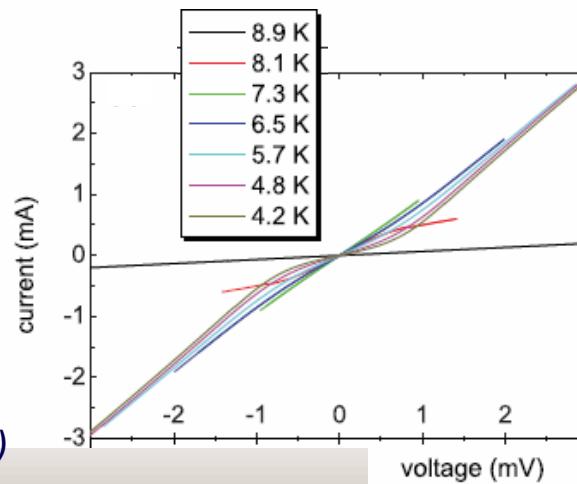
$$J_c @ 4.2 \text{ K} \sim 2.2 \times 10^6 \text{ A/cm}^2$$

Tomaschko, Leca, Selistrovski, Diebold,  
Jochum, Kleiner, Koelle, Physical Review B 85 (2012)

# Role of the epitaxial strain



Strain dependence of R vs. T

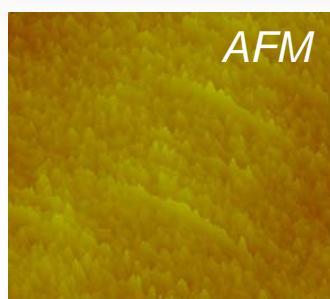


Tomaschko et al., PRB 84 (2011)

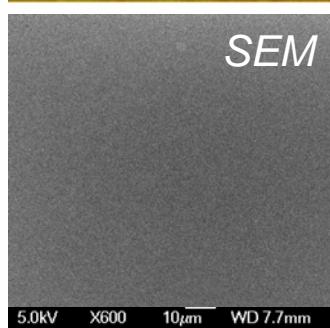
# $Sr_{1-x}La_xCuO_2$ ( $x=0.15$ ) grown on $BaTiO_3/SrTiO_3$ (001)



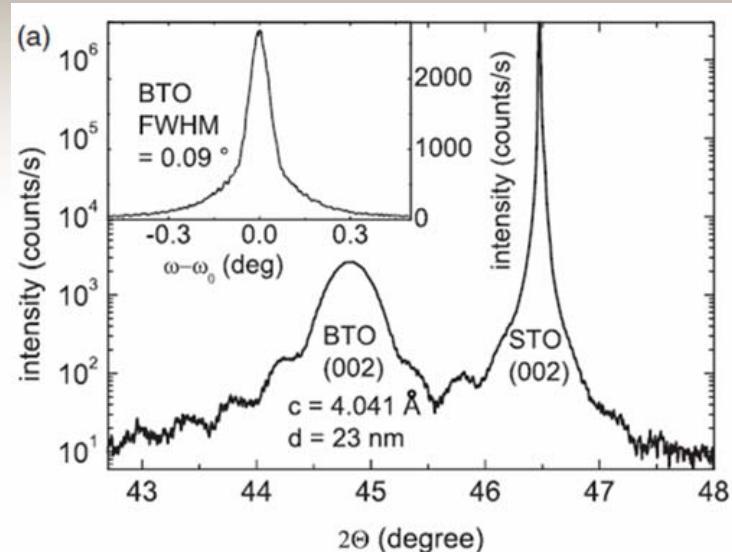
RHEED



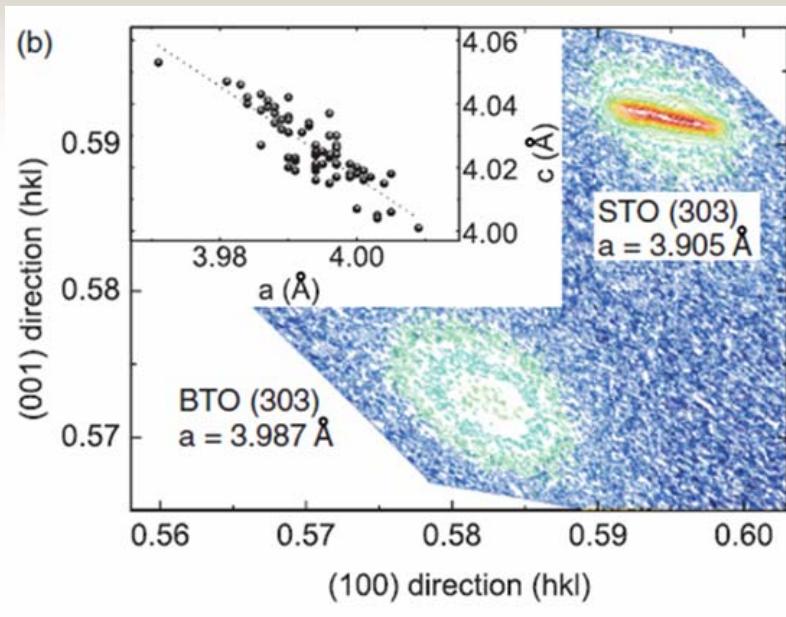
SEM



5.0kV X600 10 $\mu$ m WD 7.7mm



## The $BaTiO_3$ buffer layer

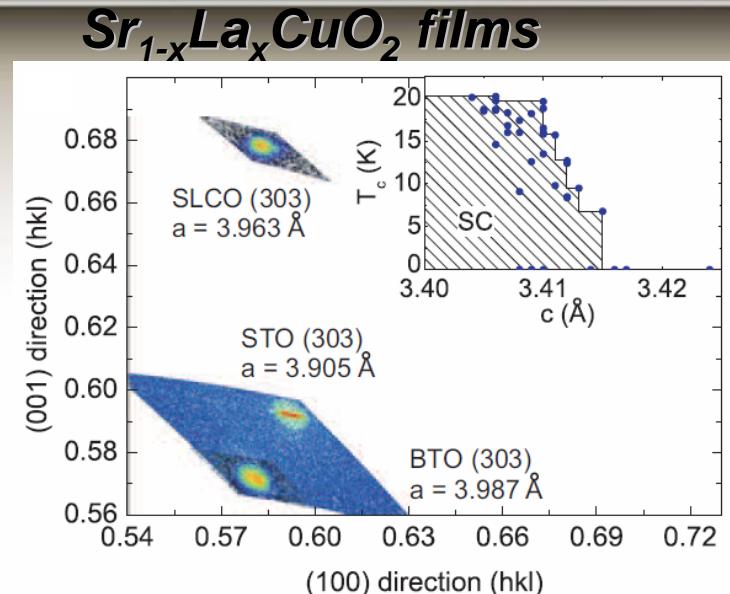
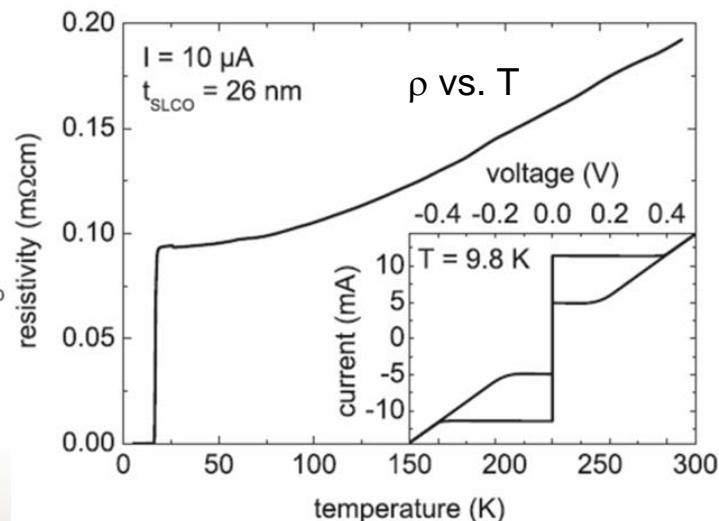
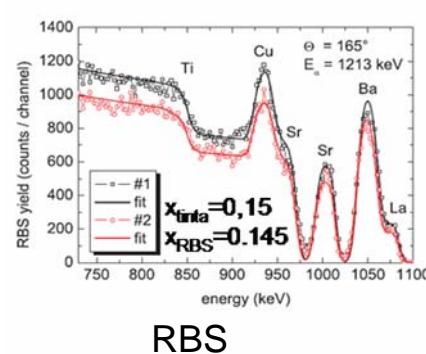
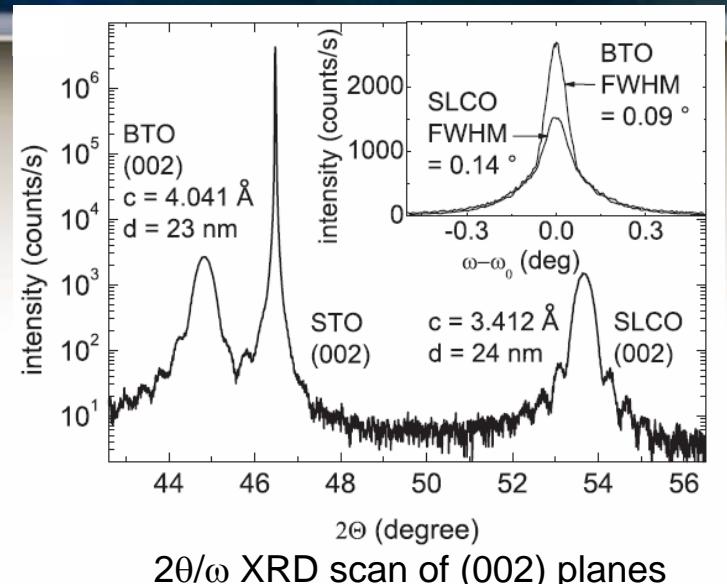
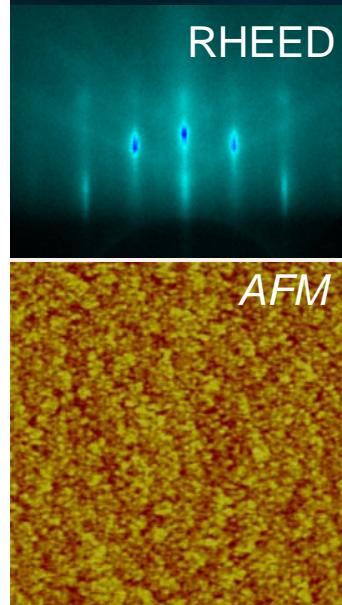


Deposition conditions for  $BaTiO_3$ :

$T_d = 750^\circ\text{C}$ ,  $P_d = 0.10 \text{ mbar } O_2$ ,  $E_d = 1.75 \text{ J/cm}^2$ ; 30 min/950°C/10<sup>-7</sup> mbar

Tomaschko, Leca, Selistrovski, Diebold, Jochum, Kleiner, Koelle, Phys. Rev. B 85 (2012)

# $Sr_{1-x}La_xCuO_2$ ( $x=0.15$ ) grown on $BaTiO_3/SrTiO_3$ (001)



XRD rsm of (303) SLCO/BTO/STO

*Deposition conditions:*

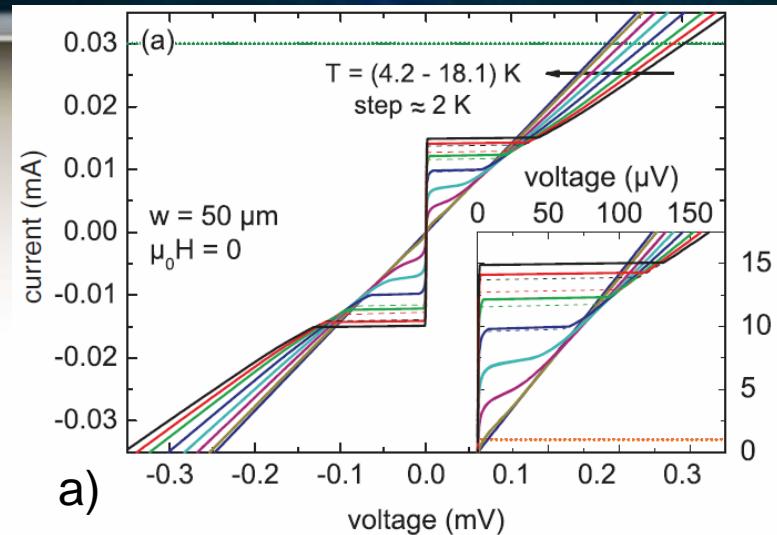
$T_d = 550^\circ\text{C}$ , 0.40 mbar  $O_2$ ,  
50 min/550°C/ $10^{-7}$  mbar

$T_{c,0} = 12-21 \text{ K}$

$J_c @ 4.2 \text{ K} = 2.1 \times 10^6 \text{ A/cm}^2$

Tomaschko, Leca, Selistrovski, Diebold,  
Jochum, Kleiner, Koelle, Phys. Rev. B 85 (2012)

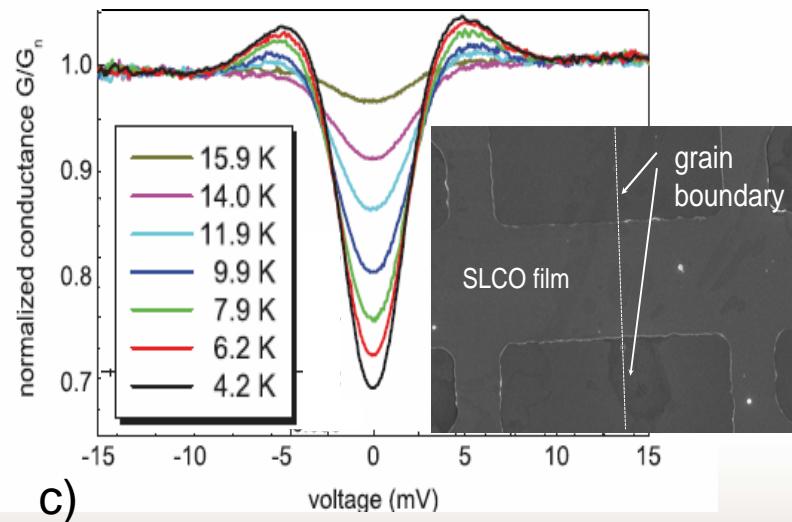
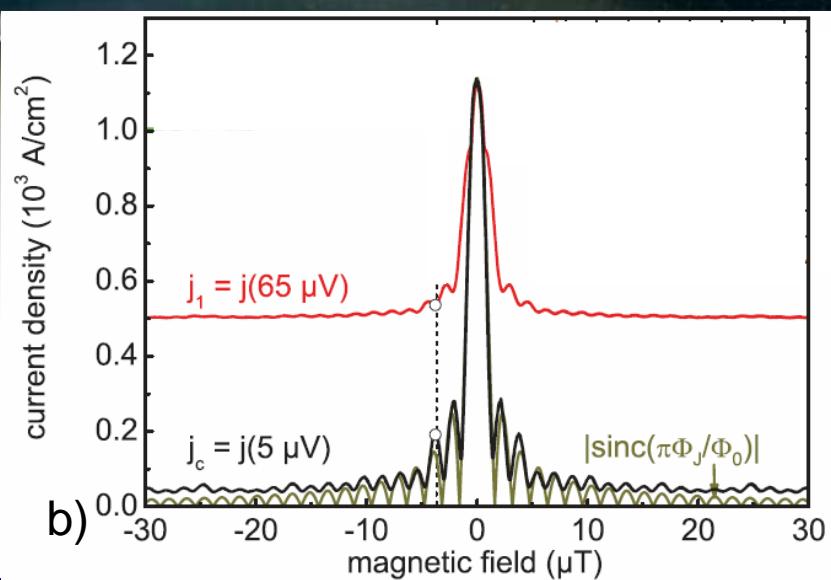
# $Sr_{1-x}La_xCuO_2$ 24° symmetric grain boundary junctions



a) ICVs: resistively and capacitively shunted junction. (RCSJ)-like, with no significant excess current;  
 $- J_c (@ 4.2 \text{ K}) \sim 1.2 \text{ kA/cm}^2$  – 1-2 orders of magnitude above  $J_c$  of 24° GB based on NCCO and LCCO

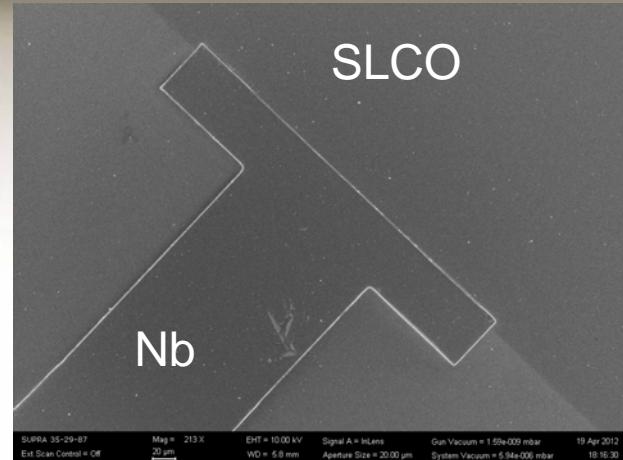
b) highly regular Fraunhofer-like patterns for different voltage criterion

c) conductance spectra did not show a zero-bias conductance peak. s-wave symmetry?  
but the V-shaped of the spectra in the subgap regime may indicate an order parameter with nodes

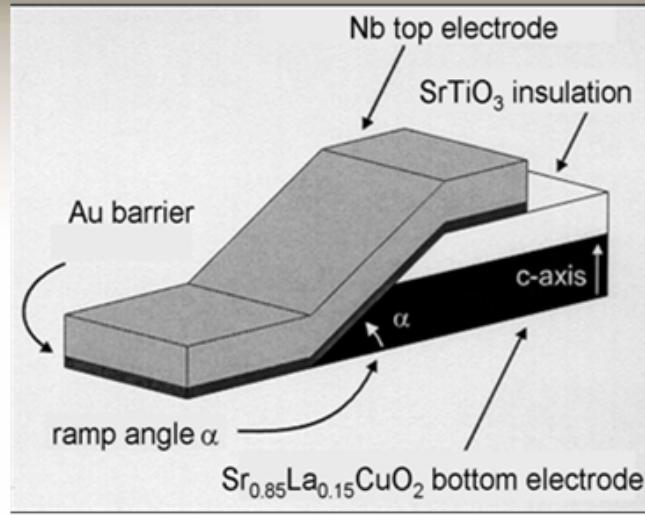


Tomaschko, Leca, Selistrovski,  
Kleiner, Koelle, Phys. Rev. B 84 (2011)

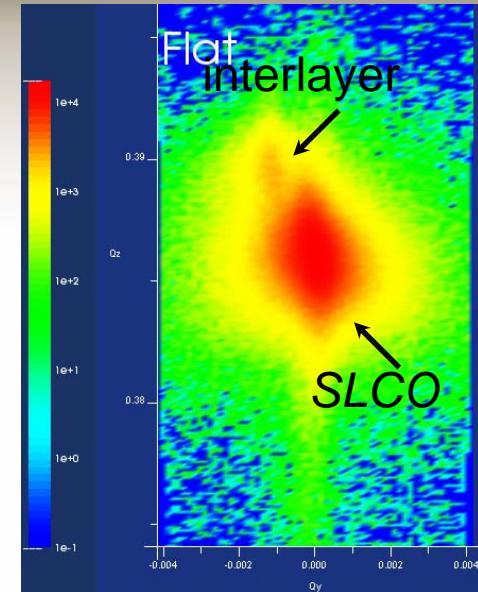
# $Sr_{1-x}La_xCuO_2$ ramp-type junctions – preliminary data



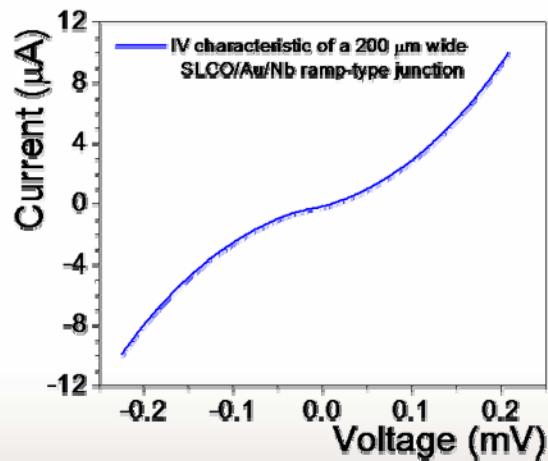
SEM image of a 200 $\mu\text{m}$  wide  
SLCO/Au/Nb ramp-type junction



Schematic representation of the  
SLCO/Au/Nb ramp-type junction  
(Smilde et al., APL 80)



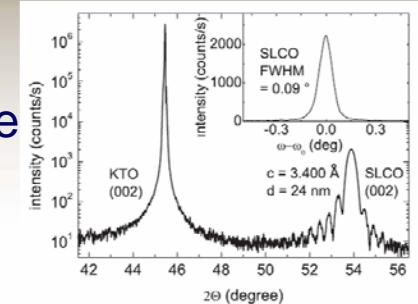
XRD rsm scan of (101) showing  
the separation of the  
SLCO and interlayer peaks



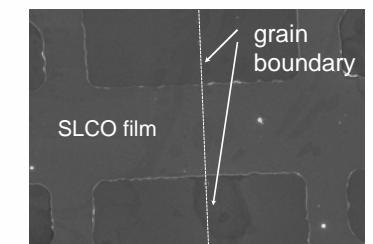
No Cooper pair tunneling observed  
- rough SLCO/Au interface

# Conclusions

- single-crystal like thin films of the electron-doped  $\text{Sr}_{1-x}\text{La}_x\text{CuO}_2$  compounds could be grown by PLD for  $x=0.15$ , but their superconducting properties (highest  $T_c$  of 25 K) are still far of those of the bulk or of the MBE grown films, but with better structural characteristics than the MBE films  $\longrightarrow$  improving  $T_c$



- epitaxial strain controls the final structural and transport properties of the films grown on  $\text{BaTiO}_3/\text{SrTiO}_3$  or  $\text{KTaO}_3$   $\longrightarrow$  new substrates (like  $\text{MeScO}_3$ )?



- first  $24^\circ$  [001] tilt grain boundary junctions were fabricated and their transport properties studied

- no transparency of the  $\text{Sr}_{1-x}\text{La}_x\text{CuO}_2/\text{Au}$  interface – no technology yet for the fabrication of the ramp-type  $\text{Sr}_{1-x}\text{La}_x\text{CuO}_2$  based junctions

